Magnetoresistance of Fe₃O₄-graphene-Fe₃O₄ junctions

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The magnetoresistance (MR) of Fe_3O_4 -graphene- Fe_3O_4 junctions has been experimentally studied at different temperatures. It is found that a barrier exists at the Fe_3O_4 /graphene interface. The existence of the interfacial barrier was further confirmed by the nonlinear I-V characteristics and nonmetallic temperature dependence of the interfacial resistance. Moreover, spin dependent transport at the interfaces contributes -1.6% MR to the whole device at room temperature and can be regulated by an external electric field. © 2011 American Institute of Physics. [doi:10.1063/1.3552679]

Graphene is a two-dimensional (2D) material with promising applications for spintronics.¹⁻⁴ The lifetime of the spinpolarized current in graphene has been demonstrated to be very long (microsecond regime), owing to the weak spin-orbit coupling of the carbon atoms.^{5–9} Due to a high carrier Fermi velocity of 10^6 m/s,¹ a large spin-relaxation length is also expected,¹⁰ which makes it a promising candidate for spin information transformation. Recently, many experiments have been carried out focusing on spin-valve devices by utilizing ferromagnetic metal as electrodes.^{3,11,12} As an alternative ferromagnetic material, ferromagnetic oxides are expected to be superior to metals as a spin injection source because of the existence of interfacial barriers between the oxide electrode and carbon material, which has the advantages including high spin polarization, environmental stability, and efficient spin injection.¹³ An important ferromagnetic oxide, Fe₃O₄, has been attracting tremendous attention over the past few decades. Fe_3O_4 is an archetype magnetic material with a high Curie temperature of about 850 K and is expected to have a fully spin-polarized electron band at the Fermi level, which is suitable for spintronic devices.^{14–17} Herein, we report on the fabrication and magnetoresistance (MR) properties of Fe₃O₄-graphene-Fe₃O₄ junctions. Our experimental results suggest that a barrier exists at the Fe_3O_4 /graphene interface. The spin dependent transport at the interfaces was found to be sensitive to the external electric field.

Graphene flakes on a Si substrate with a 300 nm SiO₂ layer were obtained by mechanical exfoliation from the Kish graphite.¹ The desired graphene flakes were selected under an optical microscope and their positions were recorded by predefined markers on the substrate. Electrode patterns on the graphene flakes were defined by electron-beam lithography. A 50 nm Fe₃O₄ thin film was grown on the substrate via oxygen plasma assisted molecular beam epitaxy.¹⁷ Finally, Fe₃O₄-graphene-Fe₃O₄ junctions were formed after lift-off process. The magnetotransport measurements were per-

formed utilizing a physical properties measurement system (Quantum Design).

The magnetization hysteresis loop of the Fe₃O₄ thin film was shown as an inset in Fig. 1(a), which indicates the ferromagnetic nature of the Fe₃O₄ thin film at room temperature. The Raman spectra of the contacted graphene flakes were measured using a Renishaw inVia Raman microscope with a 633 nm laser excitation. A typical Raman spectrum is shown in Fig. 1(a). The two peaks located at 1585 and 2647 cm⁻¹ correspond to the G band and 2D band of bilayer graphene excited by 633 nm laser, respectively.¹⁸ The background Raman signal may be due to the existence of the



FIG. 1. (Color online) (a) Raman spectrum of the graphene sample recorded with 633 nm laser excitation. Inset: hysteresis magnetization loop of the Fe_3O_4 thin film measured at 300 K. (b) The 2D band containing four components shows that the graphene is bilayer in nature. Inset: SEM image of the Fe_3O_4 -graphene- Fe_3O_4 junction.

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FIG. 2. (Color online) The MR measured at temperatures of (a) 300 K, (b) 200 K, and (c) 130 K. The red curves are measured by using contacts 1 and 2 and the black curves are measured by using contacts 2 and 3 as denoted by the SEM image in the inset of Fig. 1(a).

 Fe_3O_4 electrodes. As shown in Fig. 1(b), a good fit of the 2D band can be made with four Lorentzian peaks, which confirms the bilayer nature of the grapheme flake.¹⁸

The inset in Fig. 1(b) is a scanning electron microscope (SEM) image of the measured junction device. The upper two Fe₃O₄ electrodes (contacts 1 and 2 as denoted) contacting the graphene sheet are separated by a distance of approximately 700 nm. The lower two Fe₃O₄ electrodes (contacts 2 and 3) are connected to each other. The MR of the Fe₃O₄-graphene-Fe₃O₄ junction was measured using contact electrodes 1 and 2. However, the resistances of the Fe₃O₄ electrodes have a significant part in our two-probe devices. To decouple this contribution, the MR of the electrodes was measured using contacts 2 and 3.

The MRs, defined as MR = [R(H) - R(0)]/R(0), measured with an in-plane magnetic field, are shown in Fig. 2. More than three devices have been measured and we got the similar results from all the devices. At 300 K, the MR of the Fe₃O₄ electrodes is ~-6.8% with a 50 kOe magnetic field and no evidence of saturation was observed, which is attributed to the antiferromagnetic exchange coupling at the grain boundaries or the antiphase boundaries.^{14,17} The MR of Fe₃O₄-graphene-Fe₃O₄ junction is slightly smaller than that of the Fe₃O₄ electrodes. The graphene device consists of five distinct regions, two electrodes, two interfaces, and one graphene piece. Therefore, the MR of the device can be formulated as

$$MR = \frac{2R_F(H) + R_G(H) + 2R_{F/G}(H)}{2R_F(0) + R_G(0) + 2R_{F/G}(0)} - 1,$$
(1)

where R_F , R_G , and $R_{F/G}$ are the resistances of the Fe₃O₄ electrodes, graphene, and interface between the Fe₃O₄ and



FIG. 3. (Color online) (a) The temperature dependent resistance of the electrodes, the whole junction, and the contact interfaces. (b) The temperature dependent MR of the Fe_3O_4 /graphene interfaces at a magnetic field of 5 T.

graphene, respectively. Here, R_G may lower the MR of the device as previous studies have indicated a positive MR of graphene at high magnetic fields.¹⁹ Moreover, the interface resistances also contribute to the total resistance. At 300 K, the MR₁₋₂ (-6%) is slightly less than MR₂₋₃ (-6.8%), as seen in Fig. 2(a). By subtracting the MR contributing from the Fe₃O₄ electrodes, one can obtain that the resistance variation from the interfaces and graphene contributes -1.6%MR to the whole junction. Nevertheless, the difference of the MR between the device and the electrodes decreases with decreasing temperature, as shown in Figs. 2(b) and 2(c), which suggests that the electrode resistance may dominate the electrical properties of the device at low temperatures. Therefore, it is worth to study the temperature dependence of the resistance. Figure 3(a) shows the temperature dependent resistances of the electrodes and the graphene device measured by applying a constant current bias of 50 nA. In order to extract the interface resistance, we first calculated the activation energy of Fe_3O_4 electrode (E_{aF} =71.7 meV) by fitting the measured resistance using contacts 2 and 3 with the following formula:

$$R_{2-3}(T) = R_{F2-3} \exp(E_{aF}/k_B T).$$
(2)

 R_{F2-3} is a constant. As the resistance of the three Fe₃O₄ electrodes should have the same activation energy, the measured resistance using contacts 1 and 2 can be written as

$$R_{1-2}(T) = R_{F1-2} \exp(E_{aF}/k_B T) + R_I \exp(E_{aI}/k_B T), \qquad (3)$$

where the first term is the resistance contributing from the Fe₃O₄ electrodes and the second term is the resistance of contact interfaces. R_{F1-2} and R_I are two constants and E_{aI} is the activation energy for the Fe₃O₄/graphene interfaces. By fitting the experimental data using Eq. (3), R_I = 40872 Ω and E_{aI} =47.3 meVwere obtained. The calculated resistance of the Fe₃O₄/graphene interfaces ($R_{interfaces}$) is shown as the blue line in Fig. 3(a). $R_{interfaces}$ increased in almost two orders of magnitude as temperature decreased from 300 to 100 K. The nonmetallic temperature dependence

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FIG. 4. (Color online) (a) dI/dV vs V and I-V curves of the Fe_3O_4 -graphene- Fe_3O_4 device measured at 300 K and with magnetic field of 0 and 50 kOe. (b) The voltage dependence of the MR under a magnetic field of 50 kOe. Inset: the magnified view of MR near zero voltage.

indicates that contact barriers exist at the interfaces.

In order to extract and estimate the spin dependent transport at the interface, we define MR of the interfaces ($MR_{F/G}$) as $MR_{F/G} = R_{F/G}(H)/R_{F/G}(0) - 1$. Since the graphene resistance is on the order of k Ω which is much smaller than the interface resistance (M Ω), $R_{F/G}$ can be simply calculated by subtracting the electrodes resistance from the total resistance of the device. The temperature dependence of $MR_{F/G}$ is summarized in Fig. 3(b). It is found that $MR_{F/G}$ increases with decreasing temperature, which can be attributed to the enhanced spin polarization both in the Fe₃O₄ electrodes and graphene with decreasing temperature.

We believe that the Fe₃O₄/graphene interfaces work as a tunnel junction¹³ and the electrical behavior of the interface meets the proper tunneling criteria.²⁰ First, there is a nonmetallic temperature dependence of interface resistance as derived above [Fig. 3(a)]. Second, the device displays nonlinear I-V curves at low bias voltage, as seen in Fig. 4(a). Third, the dI/dV vs V curves [Fig. 4(a)] show a quasiparabolic relationship. Moreover, the barrier height ϕ =0.1 eV was estimated from the nonlinear I-V curves considering the electron tunneling model.²¹ Therefore, the significant change of $R_{F/G}$ under magnetic field may be attributed to the spin dependent tunneling at the interfaces.

The tunneling barriers also allow us to tune the MR of the junction by an external electrical field. Figure 4(b) shows the bias voltage dependent MR of the whole device at room temperature. The bias voltage dependent MR at 5 T can be written as

$$MR(V,5 T) = \frac{2R_F(5 T) + R_G(5 T) + 2R_{F/G}(V,5 T)}{2R_F(0) + R_G(0) + 2R_{F/G}(V,0)} - 1.$$
 (4)

As the resistances of the electrodes and graphene are independent of the bias voltage, the bias voltage dependent MR effects are due to the modification of spin dependent transport at the interfaces. At low bias voltage, the device resistance is dominated by the spin dependent tunneling at the interface. As the bias voltage is increased, the interface resistance decreases dramatically [see the inset of Fig. 4(b)], accordingly the resistance of the whole device decreases gradually, which therefore enhances the calculated MR. When further increasing the voltage, the interface tunneling less affects the electron transport and the total device resistance is dominated by the electrodes, which results in the saturation of MR at high bias voltage as shown in Fig. 4(b).

In conclusion, Fe_3O_4 -graphene- Fe_3O_4 junctions were fabricated. The nonlinear I-V characteristics and the nonmetallic temperature dependence of the Fe_3O_4 /graphene interface resistance indicate the existence of interfacial barriers. The MR of the interface increases with decreasing temperature. Our work shows that ferromagnetic oxide electrodes may be valuable for the realization of graphene-based spin devices operated at ambient temperature.

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