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# Adsorption of pure SO<sub>2</sub> on nanoscaled graphene oxide<sup>†</sup>

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Adsorption of pure SO<sub>2</sub> is studied for the first time on graphene oxide (GO). The adsorption mechanism was found to be fundamentally different from conventional carbon materials like activated carbon or carbon fibers. It was observed that GO binds SO<sub>2</sub> only by physisorption ( $\Delta H_{ads} = 18.04$  kJ mol<sup>-1</sup>) and still the adsorption capacity was found to be comparable to traditional adsorbents like activated carbon or carbon molecular sieves. The 2D layered morphology as well as the absence of extensive micropores well known to promote SO<sub>2</sub> to SO<sub>3</sub> conversion are responsible for the observed physisorption characteristics.

Sulfur compounds, especially sulfur dioxide SO2 and sulfur trioxide SO<sub>3</sub>, are found as hazardous compounds in industrial burning processes of fossil fuels e.g. coal, natural gas, oil. Even at low concentrations they are corrosive and toxic, thus presenting an immense threat to the environment. In contact with water vapor from the atmosphere they react to form sulfonic and sulfuric acid, which are the main reasons for acid rain causing extensive environmental damage. SO<sub>2</sub> removal techniques can be categorized into two broad classes namely absorption in liquids and sorption by solids.<sup>1,2</sup> Conventional absorption process like limestone scrubbing or ammonia scrubbing have some inherent disadvantages like secondary pollution, large water and energy consumption etc.<sup>1,3</sup> Adsorptive removal of SO<sub>2</sub> is a promising technique due to simplicity of operation and maintenance as well as the possibility to regenerate and fully reuse the spent adsorbent. An ideal adsorbent

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should have a high adsorption capacity at near ambient conditions, should be stable under process conditions and easily regenerable. Though zeolites and metal organic framework compounds (MOF) have a high adsorption capacity arising from their predominantly microporous framework, the presence of moisture in combination with the corrosive nature of SO<sub>2</sub> and SO<sub>3</sub> are detrimental to the long term stability of these adsorbents.<sup>4</sup> Due to its better stability under such conditions, carbon based materials especially activated carbon and carbon fibers are the most widely used adsorbents for SO2 recovery.5-9 SO2 is however known to adsorb on carbon materials both physically as well as chemically.<sup>10,11</sup> A complete desorption is typically not observed for activated carbon materials exposed to SO<sub>2</sub> even under dry conditions and low concentrations. The presence of micropores (close to 0.7 nm), humidity and oxygen in the gaseous phase is known to further facilitate chemisorption of SO<sub>2</sub>.<sup>12,13</sup> Though chemisorption promotes higher adsorption, regeneration of the adsorbent not only becomes energy intensive but also involves the consumption of the adsorbent and simultaneous generation of CO2.1

Recently, graphene oxide (GO) has attracted a great deal of interest in the field of energy and gas storage as well as catalysis.<sup>14-17</sup> This is driven by the tunability of the material in terms of its surface area, porosity and extent of oxygen surface functionalization and last but not least to its straightforward accessibility. Its mechanically stable sp<sup>2</sup> carbon framework, the presence of considerable amount of oxygen functionalities distributed along the interlayers and the possibility of largescale synthesis from inexpensive starting material qualifies GO as a plausible adsorbent material for SO<sub>2</sub> adsorption. So far, only very few studies have investigated the SO<sub>2</sub> adsorption characteristics of GO. Most of these studies are either purely theoretical or experimental works dealing with ppm concentrations of SO2 gas adsorption.18-21 However, equilibrium adsorption data of pure SO<sub>2</sub> gas is of great significance for theoretical studies on the subject as well as for predicting gas selectivity, when it comes to studying selective adsorption of gas mixtures. To the best of our knowledge no experimental

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adsorption data on GO in pure  $SO_2$  atmosphere at ambient or elevated pressures are available in the literature so far.

Herein we investigate the fundamental adsorption characteristics of GO towards pure SO<sub>2</sub>. To this end, pure SO<sub>2</sub> atmosphere rather than a flue gas composition is chosen, as it is known that the presence of oxygen and moisture influences SO<sub>2</sub> adsorption on carbon materials. Moreover, the nature of gas adsorption and hysteresis effects upon loading and deloading becomes more evident under pure gas atmosphere when both processes are unaffected by any further reaction chemistry *e.g.* with water. Furthermore CO<sub>2</sub> adsorption on GO is carried out to understand the influence of the dipole moment of both gases on the observed adsorption characteristics [ $\mu$ (SO<sub>2</sub>) = 1.6 D,  $\mu$ (CO<sub>2</sub>) = 0]. The effect of selective nitrogen functionalization on the surface of GO towards SO<sub>2</sub> and CO<sub>2</sub> adsorption is investigated as well.

For the preparation of graphene oxide (GO) (Fig. 1) graphite was first oxidized to graphite oxide according to the improved synthesis technique of Marcano *et al.* [ref. 22 and ESI<sup>†</sup> for details]. A combination of ultrasound sonication followed by various freeze-thaw cycles<sup>23</sup> was successfully employed to exfoliate graphite oxide to produce GO (see ESI<sup>†</sup> for more details). CO<sub>2</sub> adsorption measurement on this material was carried out in a modified thermogravimetric (TG) setup and high pressure SO<sub>2</sub> adsorption measurements were carried out in a commercial gravimetric setup (Rubotherm, Bochum, see ESI<sup>†</sup> for details).

 $N_2$  adsorption isotherms were measured to determine the porosity and specific surface area before and after exfoliation. A negligible amount of nitrogen is adsorbed in the as-prepared graphite oxide (see Fig. S1 in ESI<sup>†</sup>) and consequently graphite oxide exhibits a very low specific surface area of 6 m<sup>2</sup> g<sup>-1</sup>. Significant improvement in surface area is observed upon exfoliation.  $N_2$  adsorption isotherm of GO is shown in Fig. 2a. GO exhibits a type-IV isotherm in accord with previous reports<sup>17</sup>



Fig. 1 Schematic of the synthesis process of GO (synthesis and exfoliation) starting from commercial graphite.

and has a BET surface area of 268 m<sup>2</sup> g<sup>-1</sup>. Unlike activated carbons, which typically exhibits a type-I adsorption isotherm, no steep increase in adsorption is found at low pressures indicating the absence of extensive micropores. An SEM image of the GO is shown in the inset of Fig. 2a. The 2D layered morphology can be inferred from SEM and the characteristic wrinkled, layer like morphology of  $GO^{24}$  becomes evident from TEM (Fig. 2b).

Raman spectra of as-prepared graphite oxide and GO are given in Fig. 3a. The two prominent peaks are the graphitic peak (G-band) at  $\sim 1585$  cm<sup>-1</sup> and the defect induced peak (D-band) at 1350 cm<sup>-1</sup>. The G-band gives the Raman signature for all sp<sup>2</sup> carbon materials and the D-band is activated only in the presence of defects.25 The higher D-band observed in both cases is typical for graphite/graphene oxide materials and is an indicator of the structural distortions induced by the attachment of a large number of functional groups.<sup>26</sup> The  $I_D/I_G$  ratio of graphite oxide is determined to be 0.96 and after exfoliation, the ratio decreased slightly to 0.91. The small improvement in  $I_{\rm D}/I_{\rm G}$ ratio observed might be due to the removal of some functional groups during the exfoliation procedure. It is known that exfoliation leads to the breakage of hydrogen bonds between adsorbed water and surface functional groups.14 Nevertheless, significant amounts of functional groups still remain on the surface. X-ray photoelectron (XPS) survey spectrum of GO indeed confirms the presence of oxygen on the surface (see Fig. S2 and Table 1 in ESI†). To determine the nature and type of C-O bonding, the C 1s spectrum obtained under high resolution is deconvoluted into 3 peaks as shown in Fig. 3b. The main peak at 284.4 eV corresponds to the sp<sup>2</sup> bonded carbon and the other peaks are assigned to C-O (286.1 eV) and O-C=O (288.3 eV). The parameters for the asymmetric peak shape of the component at 284.4 eV are based on the peak shape originating from high purity CNT.27

 $\rm CO_2$  adsorption measurements were carried out on GO at 1 bar and 35 °C. Prior to the measurement, the sample was heated to 300 °C in an argon atmosphere (see also ESI†). The exfoliated samples adsorbed ~24.9 mg g<sup>-1</sup> of CO<sub>2</sub> at 35 °C as shown in Fig. 4a. Since the measurements are carried out on a modified TG setup, a continuous monitoring of the sample mass is possible. A significant weight loss of ~7.05% is observed already during the preliminary heating to 300 °C. In agreement with the reported TG and XPS studies,<sup>16,26</sup> the observed weight loss may be attributed to the loss of oxygen functional groups.

To investigate the effect of these weight loss on the  $CO_2$ adsorption characteristics, adsorption measurements were carried out on a sample which was heated only to 150 °C. In this case, only 2.3% decrease in weight loss was observed. However, the adsorption capacity of this sample decreased slightly to 22.4 mg g<sup>-1</sup> (Fig. 4a) as the sample activation temperature is lowered from 300 °C to 150 °C. Heat treatment of GO is known to remove surface functional groups leading to the creation of pores on the GO structure.<sup>28</sup> Increase in adsorption capacity with an increase in activation temperature is therefore in agreement with the higher porosity of the sample heated at 300 °C.

Previous studies on activated carbon<sup>29</sup> and GO<sup>30</sup> have shown that the extent of nitrogen functionalization on these materials

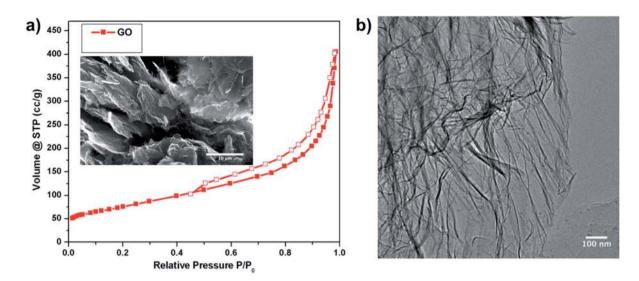


Fig. 2 (a)  $N_2$  adsorption isotherm of GO, closed symbols represents the adsorption trace, open symbols represents desorption trace. Inset in (a) shows the SEM image (scale bar = 10  $\mu$ m) of GO, (b) TEM image (scale bar = 100 nm) of GO.

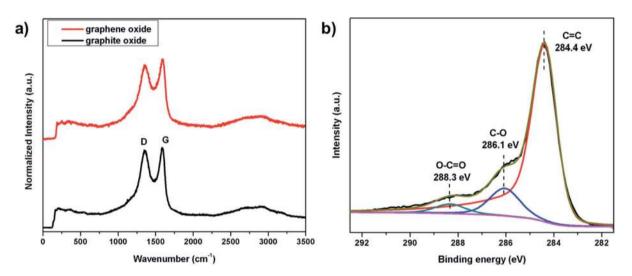


Fig. 3 (a) Raman spectra of as-prepared graphite oxide and graphene oxide (GO). (b) Experimental high resolution C 1s XPS spectrum of GO together with signal fitting.

depend on their initial oxygen surface functionalization. GO inherently has a range of various oxygen functional groups and is therefore ideally suited to introduce a significant amount of nitrogen containing surface groups by chemical modification (conversion) of initial oxygen functionalities. To determine the effect of nitrogen containing functional groups on the surface of GO on CO<sub>2</sub> and SO<sub>2</sub> gas adsorption, GO was treated with a N<sub>2</sub> rf plasma for 60 minutes which introduces a significant amount of nitrogen containing functionalities (see ESI† for more details). No morphological changes were observed in SEM upon this treatment (Fig. S3 in ESI†). N<sub>2</sub> adsorption isotherms (Fig. S4 in ESI†) of N<sub>2</sub> plasma treated GO were almost identical to that of GO (Fig. S4 in ESI†). In addition, plasma treatment did not significantly alter the specific surface of GO and was determined to be  $\sim$ 264 m<sup>2</sup> g<sup>-1</sup>. This value is only slightly lower than in the

as prepared GO material. Raman measurements of the nitrogen functionalized GO showed an increase in D-band intensity and  $I_D/I_G$  ratio increased from 0.91 to 0.97 after 60 minutes of N<sub>2</sub> plasma treatment (see Fig. S5 in ESI†).

High energy electrons and ions in an rf plasma are known to induce structural defects in the graphene lattice explaining this observed increase.<sup>31,32</sup> The XPS survey spectrum confirms the successful generation of nitrogen functionalities on GO ( $\sim$ 3.5%) by rf plasma treatment (see Fig. S6 in ESI†). Deconvolution of the high resolution C 1s spectrum (Fig. S6b in ESI†), revealed the additional presence of a new peak at 287.1 eV attributed to C=N-C groups.<sup>33</sup> Moreover, the shift of the C-O component to lower binding energies (Fig. S6c in ESI†) indicates the formation of additional C–NH. To determine the nature of the nitrogen functionalities, the high-resolution N 1s Communication

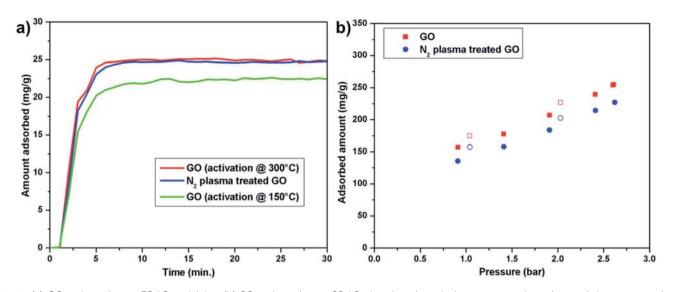


Fig. 4 (a) CO<sub>2</sub> adsorption at 35 °C and 1 bar (b) SO<sub>2</sub> adsorption at 25 °C, the closed symbols represent adsorption and the open symbol represents desorption.

spectrum is deconvoluted (Fig. S6d in ESI†) and the signals fitted corresponding (i) to C=N-C (398.7 eV) and (ii) C-NH (400.0 eV) nitrogen functionalities.<sup>34</sup>

 $CO_2$  adsorption studies on such a  $N_2$  plasma treated sample activated at 300 °C, however showed no improvement in adsorption characteristics at 1 bar and 35 °C compared to GO as shown in Fig. 4a. Tamilarasan *et al.*<sup>35</sup> have reported an enhanced  $CO_2$  adsorption for  $N_2$  plasma treated hydrogen exfoliated GO. They have observed an increase of  $\sim 2 \text{ mg g}^{-1}$  for  $N_2$  plasma treated sample over untreated GO, however at 10 °C. In the present case, the relevant measurement is carried out at 35 °C. As gas adsorption (physisorption) typically diminishes at higher temperatures, this explains the observed difference of  $CO_2$  adsorption in both experiments.

Pure SO<sub>2</sub> adsorption isotherm measurements were carried out on GO at 25 °C up to the supply pressure which is close to the saturation pressure of the adsorbate gas. The amount of pure SO<sub>2</sub> adsorbed on GO is much higher when compared to pure CO<sub>2</sub> adsorption under similar conditions of temperature and pressure (see Fig. 4b). This is attributed to the stronger van der Waals interaction resulting from the larger dipole moment of SO2 compared to CO<sub>2</sub> under these adsorption conditions. Near ambient pressure, GO has an SO<sub>2</sub> adsorption capacity of 156 mg  $g^{-1}$ . With an increase in pressure the adsorption capacity increases to 257 mg  $g^{-1}$  at 2.6 bar. GO with its moderate specific surface area of 268 m<sup>2</sup> g<sup>-1</sup> exhibits an adsorption capacity at par with other carbon materials like activated carbon, carbon molecular sieves or activated carbon fibers<sup>36,37</sup> which typically have very large specific surface areas. For example Bae et al.<sup>36</sup> observed an adsorption capacity of close to 3 mmol  $g^{-1}$  for carbon molecular sieves at around 2.6 bar while at the same pressure, GO exhibits an adsorption capacity of more than 4 mmol  $g^{-1}$ .

Desorption measurements indicated an almost complete unloading of the adsorbed  $SO_2$  with barely no hysteresis under ambient conditions. It is known from detailed studies on activated carbons and carbon fibers that SO<sub>2</sub> adsorbs on carbon materials resulting in the presence of two different adsorption regimes.<sup>10,11</sup> Weakly adsorbed SO<sub>2</sub> is the main adsorbate in regime one (adsorption energy  $\leq 50$  kJ mol<sup>-1</sup>) and strongly adsorbed SO<sub>2</sub> constitutes adsorption regime two (adsorption energy > 80 kJ mol<sup>-1</sup>). In the latter, oxidation of SO<sub>2</sub> to SO<sub>3</sub> occurs which results in a subsequently stronger adsorption by chemisorption compared to SO2 which is only physisorbed in regime one. Consequently, desorption of the more strongly adsorbing SO<sub>3</sub> requires significant higher temperatures up to 300 °C compared to that for SO<sub>2</sub>. We have carried out desorption of SO<sub>2</sub> by releasing the pressure which allows for an almost complete desorption of SO<sub>2</sub> adsorbed which is purely physisorbed on the GO surface. The 2D layered morphology as well as the absence of extensive micropores (which is known to promote SO<sub>2</sub> to SO<sub>3</sub> conversion<sup>12</sup>) is expected to be main reason for the observed physisorption characteristics.

To further confirm the predominant physisorption process of pure SO<sub>2</sub> on GO, adsorption isotherms were measured at 288 K and 308 K. As seen from Fig. 5a, the adsorption capacity decreased with an increase in temperature indicating again a physisorption mechanism. Isosteric heat of adsorption was calculated by fitting the isotherms using a virial type equation.<sup>38</sup> The heat of adsorption thus calculated is found to be 18.04 kJ mol<sup>-1</sup> at 0.5 mmol g<sup>-1</sup> loading and decreases with an increase in loading as shown in Fig. 5b (see ESI† for details). This significantly low heat of adsorption manifests clearly the physisorption nature of SO<sub>2</sub> adsorption on GO.

 $SO_2$  adsorption measurements were also performed on  $N_2$  plasma functionalized GO at 25 °C as shown in Fig. 4b. Contrary to the reports on activated carbon and activated carbon fibers,<sup>39-42</sup> no enhancement in adsorption capacity is observed after the introduction of various nitrogen functionalities. This again points to the different mechanism of  $SO_2$  adsorption on GO compared to other activated carbon materials.

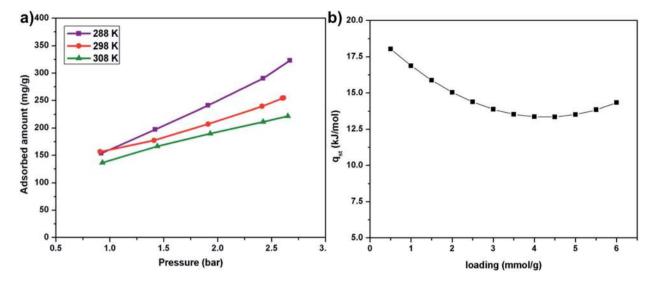


Fig. 5 (a) SO<sub>2</sub> adsorption on GO at 288, 298 and 308 K. (b) Isosteric heat of adsorption as a function of loading.

### Conclusion

CO<sub>2</sub> and SO<sub>2</sub> adsorption measurements were carried out on exfoliated graphene oxide (GO) samples. The adsorption capacity of GO is found to vary with the activation temperature. Under similar conditions of temperature and pressure, GO exhibited a greater affinity for pure SO<sub>2</sub> over pure CO<sub>2</sub>. Even with a significant amount of oxygen functionalities on the surface of GO, no chemisorption of SO<sub>2</sub> was observed. Pure SO<sub>2</sub> is found to physisorb with a low isosteric heat of adsorption of 18.04 kJ mol<sup>-1</sup> on GO and the adsorption capacity is found to be still comparable to that of other porous carbon materials. Therefore GO qualifies for swing adsorption studies of SO<sub>2</sub>. N<sub>2</sub> plasma treatment leads to ~3.5 at% nitrogen functionalization on the GO surface however, no noticeable improvement in CO<sub>2</sub> or SO<sub>2</sub> adsorption is observed with this material over native GO. The 2D layered morphology and the absence of the extensive micropores is supposed to be the reason for the observed difference from other traditional carbon materials.

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