Calculated high-pressure structural properties, lattice dynamics and quasi particle band structures of perovskite fluorides $KZnF_3$, $CsCaF_3$ and $BaLiF_3$

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Abstract

A detailed study of the high-pressure structural properties, lattice dynamics and band structures of perovskite structured fluorides $KZnF_3$, $CsCaF_3$ and $BaLiF_3$ has been carried out by means of density functional theory. The calculated structural properties including elastic constants and equation of state agree well with available experimental information. The phonon dispersion curves are in good agreement with available experimental inelastic neutron scattering data. The electronic structures of these fluorides have been calculated using the quasi particle self-consistent GW approximation. The GW calculations reveal that all the fluorides studied are wide band gap insulators, and the band gaps are significantly larger than those obtained by the standard local density approximation, thus emphasizing the importance of quasi particle corrections in perovskite fluorides.

I. INTRODUCTION

The large class of perovskite compounds, generally expressed as ABX_3 , where A and B are cations and X is an anion, have received great attention from experimentalists and theoreticians due to their interesting properties from the fundamental physics and chemistry point of view. They possess a wide range of applications such as lenses without birefringence, and exhibit magnetism, pizeoelectrics, ferroelectrics etc.¹ The physical properties of perovskites may also have implications towards understanding the Earth's lower mantle.^{2,3} Fluoride perovskites is one particular subclass, which has found important technological applications in the field of optics. The main advantage is that these materials can be used as light emitting materials in the deep ultraviolet regime.⁴ In the present work the three technologically important fluoroperovskite KZnF₃, CsCaF₃ and BaLiF₃ are considered. They all crystallize in the cubic perovskite structure. $KZnF_3$ is a promising candidate for application as a radiation detector.⁵ Several experimental studies focused on synthesis, lattice dynamics, high-pressure structural stability, elastic constants, optical absorbtion and photoluminescence.^{6–12} In addition, several theoretical studies of KZnF₃ investigated its electronic structure and optical properties, lattice dynamics and thermodynamic properties, crystal fields and influence of 3d transition metals dopants.^{13–16} Similar to KZnF₃, the CsCaF₃ perovskite also was investigated in several experimental and theoretical studies addressing the electronic structure and optical properties, low energy phonon dispersion curves, green luminescence upon Eu²⁺ doping, thermodynamics at low temperatures, and the equation of state.^{8,17-23}

BaLiF₃ has been of interest for application within lithography due to the short wavelength absorption edge at 123 nm (10.1 eV).²⁴ In addition, BaLiF₃ may also be used as an effective dopant to alter the dielectric properties of BaTiO₃.²⁵ The mechanism of ionic conductivity has also been investigated in BaLiF₃ with the purpose of enhancing the ionic conductivity by means of intrinsic defects.²⁶

The infrared dielectric dispersion of BaLiF₃ measured at several temperatures confirmed that this compound is one of the most stable fluoroperovskites.²⁷ This was further confirmed by inelastic neutron scattering measurements of the phonon spectrum.²⁸ The electronphonon coupling in Ni²⁺ doped BaLiF₃ has been measured and related to the change in bonding as manifested in the computed vibronic spectra.²⁹ The luminescence of BaLiF₃:Eu²⁺ has been investigated under pressure.³⁰ High pressure studies on fluoride compounds has resulted in various interesting physical properties. A recent high pressure experimental and theoretical studies on fluoride compound CaF_2 reported superionic behavior even at ambient conditions which decreases with compression.³¹ Recently, the structural stability of cubic BaLiF₃ under pressure has been confirmed experimentally up to 50 GPa.³² Similar results have been reported recently for the perovskite fluorides KMgF₃ and CsCdF₃ using high-pressure experimental techniques up to 40 and 60 GPa, respectively.^{33,34} *i.e.*, the cubic phases of KMgF₃ and CsCdF₃ are found to be stable in the pressure ranges studied.

The present paper reports the results of a systematic and comparative study of the structural properites, lattice dynamics and electronic structures, of the three perovskite fluorides $KZnF_3$, $BaLiF_3$ and $CsCsF_3$. The remainder of the paper is organised as follows. Section II describes the computational details. The calculated ground state properties, elastic constants, and phonon dispersion curves are discussed in section III, and section IV presents the quasiparticle band structures of the fluoroperovskite compounds. Conclusions of the work are presented in section V.

II. COMPUTATIONAL DETAILS

Three complementary density functional tools have been employed to study the ground state and elastic properties, lattice dynamics, and quasiparticle band structures of KZnF₃, CsCaF₃ and BaLiF₃. The structural and elastic properties were calculated using the allelectron linear muffin-tin orbital method³⁵ in the full-potential implementation of Ref. 36. In this method the crystal volume is divided into two regions: nonoverlapping muffin-tin spheres surrounding each atom and the interstitial region between the spheres. We used a double κ spdf LMTO basis set to describe the valence bands. The calculations include 3*s*, 3*p*, 4*s*, 4*p* and 3*d* partial waves for potassium, 4*s*, 4*p* and 3*d* partial waves for zinc, 5*s*, 5*p*, 6*s*, 5*d* and 5*d* partial waves for Ba, and 2*s* and 2*p* partial waves for fluorine. The exchange correlation potential was calculated within the local density approximation (LDA)³⁷ as well as the generalized gradient approximation (GGA) scheme.³⁸ The charge density and electron potential inside the muffin-tin spheres were expanded in terms of spherical harmonics up to $l_{max}=6$, while in the interstitial region, these quantities were expanded in plane waves, with 28670 waves (energy up to 194.81 Ry) being included in the calculation. Total energies were calculated as a function of volume, for a $(16 \times 16 \times 16)$ k-mesh, corresponding to 165 k-vectors in the irreducible wedge of the Brillouin zone (BZ), and the results fitted to the Birch equation of state³⁹ to obtain the ground state properties. The elastic constants were obtained from the variation of the total energy under volume-conserving strains, as outlined in Ref. 40.

The phonon frequencies were calculated based on the density functional linear-response approach⁴¹ within the GGA,³⁸ combined with the plane-wave pseudopotential method as implemented in the PWSCF package.⁴² Norm-conserving pseudopotentials were used to describe the valence electrons. The BZ integrations were carried out with a $(6 \times 6 \times 6)$ Monkhorst-Pack (MP) grid.⁴³ Kinetic energy cutoffs of 90 Ry, 90 Ry and 120 Ry were used for CsCaF₃, BaLiF₃ and KZnF₃, respectively. A $(4 \times 4 \times 4)$ q mesh in the first BZ was used for the interpolation of the force constants needed for the phonon dispersion curve calculations. The theoretical equilibrium lattice constant of 4.096 Å, 4.604 Å, and 4.085 Å for KZnF₃, CsCaF₃ and BaLiF₃, respectively, are obtained by total energy minimization and used in the phonon calculations.

The electronic structures of $KZnF_3$, $CsCaF_3$ and $BaLiF_3$ have been calculated with the quasiparticle self-consistent GW (QSGW) (where G denotes the Greens function and W denotes the screened Coulomb interaction) approximation,^{44,45} in the FP-LMTO implementation of Ref. 46. This calculation also included a double- κ LMTO basis set ($\ell_{max} = 6$), however additionally including the F 3s and 3p partial waves (treated as local orbitals⁴⁶) for better description of the unoccupied band states. An $(8 \times 8 \times 8)$ MP k-mesh was used for the calculation of the screened interaction, which is evaluated in the random-phase approximation. Convergence tests showed only minor changes compared to a $(6 \times 6 \times 6)$ mesh. The most common type of GW calculations use a selfconsistent LDA or GGA band structure as input for the evaluation of the G and W operators. The special feature of the quasiparticle self-consistency in the QSGW method is that the one-particle band structure, which is used as input for the evaluation of the Greens' function, is iterated in a further selfconsistency loop so as to come as close as possible to the output GW band structure.^{44,45} Though appealing from a theoretical point of view, this procedure has a tendency to slightly overestimate semiconductor gaps,⁴⁷ a fact traceable to the neglect of vertex corrections. As a simple remedy of this, Chantis et al.⁴⁷ suggested the hybrid QSGW (h-QSGW), by which the LDA and QSGW self energies are mixed in the proportions 20% - 80%. See Ref. 48 for a detailed description. Here we also apply the h-QSGW method to the fluorite perovskites to monitor its effect for this class of wide-gap insulators

III. GROUND STATE, ELASTIC PROPERTIES AND PHONON DISPERSIONS

The calculated structural properties such as lattice parameters, and bulk moduli of KZnF₃, CsCaF₃ and BaLiF₃ are presented in Table I along with available experimentalal information and results of other calculations. In the case of KZnF₃ the calculated lattice parameter within LDA is 0.9% lower when compared with the experimental value and the corresponding bulk modulus is 23% overestimated. In contrast, the calculated lattice parameter using GGA overestimates the experimental value by 2.4%, and the bulk modulus is 6.8% too low. This is the usual trend observed in the LDA and GGA schemes. In the case of $CsCaF_3$ the LDA calculated lattice constant is lower by 2.8% when compared to experiments, and the corresponding bulk modulus is 44% higher. Similarly, the GGA lattice parameter is 0.4% lower and the corresponding bulk modulus is 8% too large. A similar situation is seen in the case of $BaLiF_3$. The calculated bulk modulus is in excellent agreement with the two reported experimental values, which is, however, a bit fortuitous. Since the calculated equilibrium volume is overestimated (by about 0.8%) with GGA (and underestimated by about 2.6 % with LDA), an error solely depending on the error in volume is introduced in the calculated bulk modulus. Therefore we recalculated the bulk modulus also at the experimental volume in a manner similar to our earlier work³³ (see Table I). This diminishes the discrepancies between the LDA and GGA results, as expected. In addition, the LDA bulk modulus now becomes *smaller* than the GGA for all the three compounds, and both functionals are seen to actually overestimate the bulk modulus. For $KZnF_3$ two experimental determinations of the bulk modulus are at variance. In particular the high value for the pressure derivative of the bulk modulus $(B'_0 = 11)$ reported by Ref. 11 is at odds with the general range $B'_0 \sim 4-5$ found for the fluoride perovskites by both experiments and theory. The authors suggest that the limited pressure range covered in their study may be the cause of their high fit value for B'_0 . In Figure 1 the GGA and LDA equations of state of KZnF₃, CsCaF₃ and BaLiF₃ in the pressure range from 0 to 50 GPa are compared with the available experimental data.^{11,32} The agreement between theory and experiment is excellent for $BaLiF_3$ in Fig. 1(c), while the agreement is less satisfactory for $KZnF_3$ in Figure 1(a). A special feature in $KZnF_3$ is the role of the Zn 3d states, which occur as a narrow resonance within the F valence bands reflecting their semi-localized character. Their bonding properties are usually ill described in LDA/GGA and this may be the cause of the discrepancy here.

The elastic constants of KZnF₃, CsCaF₃ and BaLiF₃ are presented in Table II along with the experimental values and other theoretical values. The LDA overestimates all the C_{11} , C_{12} and C_{44} by between 10% and 25%, while the elastic constants obtained within GGA are much closer to the experimental values. As for the bulk modulus the elastic constants also depend sensitively on the volume, and therefore, the same argument as for the bulk modulus can be applied here. We have however, refrained from recalculating all the elastic constants with volume correction, but wish to mention that the excellent agreement between experiment and the GGA elastic constants should be interpreted with some care. Another point of caution is the fact that the calculated values pertain to 0 K, while experiments are performed at room temperature. An increase in temperature generally tends to reduce the elastic constants because of thermal expansion.

The calculated phonon dispersion curves at the optimized theoretical lattice parameters are shown in Figure 2. The available experimental inelastic neutron scattering frequencies^{8,28} are included and good agreement between theory and experiment is found. The dynamical stability of the cubic structures is confirmed by the absence of any imaginary modes in the calculated dispersions. There are 15 phonon branches in the full phonon dispersion since the unit cell consists of five atoms which give rise to three acoustic and twelve optical phonon branches. The agreement with experiment is excellent for $BaLiF_3$, and for the few frequencies determined for $CsCaF_3$, while there is some discrepancy to be seen for $KZnF_3$ around the lowest frequencies at the M and R points. The calculations find a noticeable gap between the lowest optical mode and the highest acoustic mode in the calculations for $BaLiF_3$ and $CsCaF_3$, which does not appear in $KZnF_3$. In addition, in $CsCaF_3$ a large second gap appear at high frequency. The experiments have only managed to map the lowest frequencies and thus cannot confirm or discard this prediction. Evidently, it would be of great interest to obtain experimental phonon frequencies in the higher end of the spectrum for all three compounds, in particular to test the theoretical prediction that the phonon spectrum for CsCaF₃, BaLiF₃ and KZnF₃ contain two, one and zero gaps, respectively.

The phonon partial densities of states are also included to the right of Figures 2(a)-(c). The acoustic modes in BaLiF₃ and CsCaF₃ are dominated by the motion of the heavy Ba and Cs, respectively, while in KZnF₃ the atomic masses are more similar, and the atomic vibrations are fully coupled to each other as shown in figure 2(a). Furthermore, the acoustic band width is smaller in CsCaF₃ than in BaLiF₃, which reflects the larger lattice constant of CsCaF₃, while the second gap (between 8 and 11 THz) in the phonon spectrum of CsCaF₃, which does not find a counterpart in BaLiF₃, reflects the different masses of Ca and Li. Thus, there is a significant Li component to the highest frequencies in BaLiF₃, while no Ca weight is found in the highest CsCaF₃ frequency range.

IV. QUASIPARTICLE BAND STRUCTURES

The QSGW band structures of KZnF₃, CsCaF₃ and BaLiF₃ are presented in Figure 3. All three compounds exhibit large gaps, of 10.0 eV, 11.8 eV and 11.8 eV, respectively, which are significantly larger than the gaps calculated within the LDA (Table III). The experimental gap has only been reported for the case of BaLiF₃ to be 8.4 eV,⁴ which is 3.4 eV lower than the calculated QSGW value. The h-QSGW reduces the calculated gaps by about 1 eV (Table III), i. e. this can not account for the discrepancy. There exists the possibility that the ideal perovskite structure assumed in the calculations is not the same as that of the crystals grown in the laboratory, which may be hampered by intrinsic defects etc.

The conduction band minimimum occurs in all cases at the Γ point, while the valence band maximum (VBM) also falls at Γ for BaLiF₃, *i.e.*, this is a direct gap insulator. The three top valence states at Γ , M and R are rather close, with M (R) at -0.17 eV (-0.20 eV) relative to the VBM in BaLiF₃. In contrast, for CsCaF₃ and KZnF₃ the gap is indirect. In CsCaF₃ the VBM occurs at the M point while the topmost valence state at Γ is about 0.25 eV lower in energy. In KZnF₃ the VBM occurs on the line connecting the M and R points, and the topmost valence state at Γ is 1.5 eV lower in energy.

The F band widths are 4.0 eV, 2.1 eV and 3.2 eV in $KZnF_3$, $CsCaF_3$ and $BaLiF_3$, respectively. The low band width of $CsCaF_3$ reflects its larger lattice constant. $KZnF_3$ and $BaLiF_3$ have similar lattice constants, however in $KZnF_3$ the Zn 3d bands falls at the bottom of the F bands partially hybridizing and thus effectively enhancing the F band width. The semicore states are slightly shifted towards higher binding energies by the GW approximation compared to their LDA positions, by about 2 eV for the K 3p states of KZnF₃ (located at 12 eV below VBM), by about 0.1 eV for the Cs 5p states in CsCaF₃ (located at 4.5 eV below VBM), and by about 0.6 eV for the Ba 5p states of BaLiF₃ (located at 9 eV below VBM). Thus the most significant effect of the GW approximation is the more or less rigid shift of the conduction bands with respect to the valence bands, which is of the order of 5-7 eV for the three fluorides considered.

V. CONCLUSIONS

The electronic structures and vibrational properties of the KZnF₃, CsCaF₃ and BaLiF₃ perovskite fluorides have been investigated with density functional methods. The calculated phonon frequencies are in good agreement with experimental values, which however are limited to the lowest part of the spectrum, and a full experimental mapping of the phonon dispersion curves is desirable. The calculated pV-curve of BaLiF₃ is in excellent agreement with experiment, while the agreement between theory and experiment is less satisfactory for KZnF₃, and no experimental data exist for CsCaF₃.

The QSGW quasiparticle band structures reveal significant insulating gaps in the fluorides studied, which are about 6 eV larger than predicted by the LDA/GGA. Most of this effect may be seen as a rigid shift of the conduction bands with respect to the valence bands. The F valence bands are only slightly wider in QSGW than in LDA, while some shifts towards higher binding energies are found for the secondary (semi-core) valence states. All of these effects are usual trends of the GW method, which are also observed in comparison of Hartree-Fock and LDA band structures, hence they may be ascribed to the non-local character of the electron-electron interaction.

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TABLE I. Calculated Lattice constants (in Å), bulk moduli B_0 (in GPa) and its pressure derivatives B'_0 , of KZnF₃, CsCaF₃ and BaiF₃. The bulk moduli have been calculated both at the experimental and theoretical volumes ($B_0(V_0^{exp})$ and $B_0(V_0^{th})$, respectively). The experimental bulk moduli reported for KZnF₃, CsCaF₃ and BaLiF₃ are obtained from high pressure structural measurements (Refs. 11 and 32), or derived from the elastic constants: $B = (C_{11} + 2C_{12})/3$ (Refs. 49, 55 and 28).

Compounds	Lattice constant	$B_0(V_0^{\mathrm{th}})$	$B_0(V_0^{\exp})$	B'_0		
KZnF ₃						
LDA^{a}	4.017	95.2	83.5	4.4		
GGA^a	4.154	72.3	99.8	4.3		
Expt.	$4.054^b, 4.060^c$		$77.6^b, 54.8^c$	11^c		
Other theory	$4.021^d, 4.072^e, 4.07^f,$	89.2^{d}		4.51^{d}		
	$3.796^g, 3.897^h, 4.1499^i$	66.4^{i}		4.22^{i}		
$CsCaF_3$						
LDA^{a}	4.401	73.5	47.8	4.6		
GGA^a	4.545	55	58.2	4.5		
Expt.	4.526^{j} ,		50.9^j			
Other theory	$4.569^k, 4.579^l, 4.5885^m$	$50.2^k, 50.9^l, 45.8^m$		$4.27^k, 4.6^m, \! 3.8^l$		
D-I :E						
BaLiF ₃	0.000					
LDA^{a}	3.892	98.2	65.8	4.6		
GGA^a	4.028	75.6	80.7	4.5		
Expt.	3.995^{n} ,		$79(3)^n, 75.9 \pm 1.3^o$	$5.35 {\pm} 0.15^{o}$		
Other theory	4.050^{p}	64.5^{p}		4.6^{p}		
^{<i>a</i>} Present work; ^{<i>b</i>} Ref. 49; ^{<i>c</i>} Ref. 11; ^{<i>d</i>} Ref. 13; ^{<i>e</i>} Ref. 50; ^{<i>f</i>} Ref. 51; ^{<i>g</i>} Ref. 52; ^{<i>h</i>} Ref. 53; ^{<i>i</i>} Ref. 54;						

 ${}^{j}{\rm Ref.}$ 55; ${}^{k}{\rm Ref.}$ 23; ${}^{l}{\rm Ref.}$ 22; ${}^{m}{\rm Ref.}$ 21; ${}^{n}{\rm Ref.}$ 28; ${}^{o}{\rm Ref.}$ 32; ${}^{p}{\rm Ref.}$ 56.

Compounds	C_{11}	C_{12}	C_{44}	G				
$KZnF_3$								
LDA	175.7	55.0	47.2	52.5	Present			
GGA	134.8	41.1	44.1	45.2	Present			
GGA	99.2	38.0	29.1	29.7	Ref. 54			
GGA	111.8	49.8	31.4	31.2	Ref. 16			
Expt.	134.5 ± 1	$52.7 {\pm} 0.5$	$38.1 {\pm} 0.2$	39.2	Ref. 49			
Expt.	146	54	39		Ref. 10			
$CsCaF_3$								
LDA	166.2	27.3	29.2	45.3	Present			
GGA	121.0	22.1	29.4	37.4	Present			
GGA	98.5	24.8	27.6	31.0	Ref. 23			
GGA	99.9	23.6	25.0	29.6	Ref. 22			
Expt.	102 ± 1	$25.3 {\pm} 0.5$	$25.5 {\pm} 0.5$	30.6	Ref. 55			
$BaLiF_3$								
LDA	232.3	31.2	42.2	65.5	Present			
LDA	163.8	50.8	50.0		Ref. 56			
GGA	149.2	33.1	58.0	58.0	Present			
Expt.	130±1	46.5 ± 0.5	48.7 ± 0.5	45.9	Ref. 28			

TABLE II. Calculated elastic constants and shear modulus (G), all expressed in GPa, for $KZnF_3$, $CsCaF_3$ and $BaLiF_3$ at the theoretical equilibrium volume.

	•			
compound	LDA	QSGW	h-QSGW	Expt.
$KZnF_3$	2.2	10.0	8.7	
$CsCaF_3$	6.1	11.8	10.7	
$BaLiF_3$	6.3	11.8	10.7	8.41^{a}

TABLE III. Comparison of the fundamental band gap (in eV), for $KZnF_3$, $CsCaF_3$ and $BaLiF_3$, as calculated with GGA and QSGW.

^{*a*}Ref. $\overline{4}$;



FIG. 1. Calculated equation of state (LDA and GGA) versus experimental pV relations for (a) KZnF₃, (b) CsCaF₃ and (c) BaLiF₃. Experimental data are from Refs. 11 (KZnF₃) and 32 (BaLiF₃).



FIG. 2. Calculated phonon dispersion curves of (a) $KZnF_3$ and (b) $CsCaF_3$ and (c) $BaLiF_3$ at the theoretical lattice parameter. The experimental data are shown by dots and taken from: $KZnF_3$: (Ref. 8); $CsCaF_3$: (Ref. 8); $BaLiF_3$: (Ref. 28).



FIG. 3. (Color Online) Calculated LDA (red (gray thin) lines) and QSGW (black thick lines) band structures of (a) KZnF₃, (b) CsCaF₃, and (c) BaLiF₃ at the experimental lattice constants. The zero of energy is situated at the valence band maximum. The BZ special points are Γ : (0,0,0); X: (1,0,0); M: (1,1,0) and R: (1,1,1) in units of $2\pi/a$.